2CuMn₂O₄ nanostructures: Facial synthesis, structural, magnetical, electrical characterization and activation energy calculation

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ABSTRACT: The work is the report about stearic acid sol-gel synthesis method, magnetically, electrical characterization and activation energy of copper manganese oxide nanostructures. The CuMn₂O₄ nanostructures are synthesized at a temperature of 600°C using the sol-gel method. The structural analysis using X-ray diffraction (XRD) and Scherrer equation show that the crystallite size of CuMn₂O₄ is about 23.0 nm. The activation energy value is measured by temperature-programmed reduction (TPR) analysis at three heating rates of 10, 15 and 20 $^{\circ}$ C/min and Kissinger equation The Ea value is 28.1 kJ/mol. Vibrating sample magnetometer (VSM) analysis of the nanostructures indicates moderate ferromagnetic properties. Examination of the dielectric properties by the LCR meter indicates the semiconducting property of the nanostructure. The highest conductivity increase was in the frequency of 308 MHz equal to 70.7 µS. Therefore, the CuMn₂O₄ nanostructures are potential candidates in fuel cells, telephones, loudspeakers and transformers due to their properties and convenience, non-toxic and environmentally friendly production methods.

Keywords: Activation Energy; CuMn2O4 Nanostructures; Electrical Properties; Ferromagnetism; Kissinger Equation.

INTRODUCTION

Knowledge of solid-state materials in chemistry is essential to understand the importance of many advanced ties of advanced nano-materials, their ultra-small size, materials $[1,2]$. The unique physicochemical properhigh reactivity and large surface area-to-mass ratio, have made them distinct from other materials [3]. The general formula of spinel oxides is as AB_2O_4 [4] struc tures with this formula show a variety of applications in electronics, magnetic materials and catalysts [5]. In $CuMn₂O₄$ normal spinel, the electron configuration of Mn^{3+} is as 3d⁴ and the electron configuration of Cu^{2+}

is $3d^9$, which occupy the tetrahedral and octahedral graded tetragonal-spinel structure using a single-step sites, respectively. The researchers have prepared a demethod, and measured the magnetization with respect to temperature and magnetic field $[6]$. In addition, they synthesized CuMn₂O₄ nanopowders using a low-tem perature mixing technique; the mean size of the syn-
thesized nanostructures was about 50 nm [7]. Recently, synthesized CuMn₂O₄ nanopowders using a low-tem-
perature-mixing technique; the mean size of the synthe structure, morphology and magnetic properties of CuMn₂O₄ have been studied [8, 9]. CuMn₂O₄ spinel ide fuel cell and compared to layered compounds such has been used as a protective coating on a solid oxas $LiNiO₂$ and $CuCO₂$ [10]. In addition, the CuMn₂O₄

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spinel has an economic and environmental advantage as a fuel cell process [11]. The CuMn₂O₄ nanostructures have semiconducting properties and can be used in the electronics industry. There are several methods for preparation of nanostructures such as combustion method $[12]$, melting technique $[13]$, chemical vapor deposition [14], atomic layer deposition [15], sol-gel [16], hydrothermal [17] and ceramic method $[18]$.

The sol-gel method was first used historically for ly used for different fields such as glasses, ceramics, building and decorative materials, and then was widecatalysts, coatings, composites and fibers in the last century [19]. The sol-gel process has been applied as pounds and chemical materials. This method contains a versatile method for preparation of inorganic comadvantages of easy components adjustment, chemical homogeneity, low cost and low calcination tempera-
ture [20].

Preparation of CuMn₂O₄ nanostructures by solgel method was the most important challenge in this cination at 600° C. Various methods containing X-ray study. The nanopowders were synthesized after caltion (TPR), vibrating-sample-magnetometer (VSM) diffraction (XRD), temperature-programmed reducvation energy, kinetic properties, magnetic behavior tures. Using the techniques structural properties, actiand LCR meter were used for study of the nanostrucand electrical properties of $CuMn₂O₄$ nanostructures were investigated.

EXPERIMENTAL

MATERIALS AND METHOD

In the present study, the CuMn₂O₄ nanostructures were prepared by sol-gel method using stearic acid as complex agent, copper and manganese acetate as metal cation sources. The metal cations penetrate from ing point of stearic acid. First, 0.3 mol stearic acid the aqueous phase to the organic phase over the melt-.(97% purity, Merck) was melted in a beaker at 73° C. etate (99.9% purity, Aldrich) and 0.1 mol of Mn (II) Then, in a stoichiometric ratio of 0.5 mol Cu (II) acter and added to molten stearic acid. At this stage, the acetate (99.9% purity, Aldrich) were dissolved in watwo soluble phases became a gel. Finally, the resulting

Scheme 1. Schematic illustration of CuMn₂O₄ nanostruc-
tures synthesis steps.

gel contained Cu (II) and Mn (II) ions. The gel was heated to boiling point of stearic acid at 100 to 350° C for 1 h until an aqueous phase was formed. Then, the cination process. Finally, the material was ground in a resulting material was heated to 600° C for 4 h for calmortar and prepared to identify its structure and properties.

CHARACTERIZATION

 XRD pattern of $CuMn₂O₄$ nanopowders was evaluated with PTF 3003 model from SEIFERT spectrometer using Cu-Kα beam $(\lambda = 1.418$ Å, 30 kV and 30 mA) in range of ($2\theta = 10^{\circ}$ - 80°) to determine structure and size of the nanostructures (JCDPS 74-1921). Typical degradation processes were investigated using TPR technique by a thermal conductivity detector (TCD) of a gas chromatograph (6890 plus, Toseye Hesgarsazan Asia Co., Iran). The kinetic parameters were studied using adsorption/desorption isotherms of N_2 at 196 °C using Chem-BET Pulsar TPR/TPD/BET (Toseye Hes-
garsazan Asia Co., Iran).

RESULT AND DISCUSSION

2CuMn of analysis Structural 4O nanostructures

 XRD pattern (Fig. 1) showed the cubic structure of CuMn₂O₄ nanopowders with space group */Fd-3 m.

Fig. 1. XRD pattern of CuMn₂O₄ nanostructures calcined at .equation Scherrer Equation Scherrer equation Scherrer equation Scherrer equation.

The highest peak is at a diffraction angle of 36.1°. Several-other tall peaks were also observed which agree with the diffraction pattern of the CuMn₂O₄ reference sample. As shown in Fig. 1, the characteristic peaks of CuMn₂O₄ are at the 20 angles of 18.63 °, 30.69 °, 32.68°, 36.1°, 37.69°, 43.88°, 54.26°, 57.91°, 60.23 \degree , 63.54 \degree , 75.10 \degree , respectively. The mean crystallite size of $CuMn₂O₄$ nanostructures was calculated using XRD pattern and Scherrer equation (Eq. 1):

$$
D = \frac{K\lambda}{B\cos\theta} \tag{1}
$$

where, D is the mean size of nanostructures, $K = 0.9$ is tures, $λ$ is the wavelength of 0.15418 nm, $β$ is the full a crystallite shape factor for the spherical nanostrucwidth at half the maximum and θ is the peak position.

The graph of $1/\beta$ on the X-axis and cos θ on the Y axis is shown in Fig. 2. Using the slope of the line of this equation, the mean size of the nanostructures was about 23.03 nm.

Evaluation of activation energy and kinetic proper-
ties

Thermal properties, either thermal degradation or calculation of the activation energy of the degradation process was favorite for estimation the thermal properties of $CuMn₂O₄$ nanostructures. The activation energy of degradation (Ed) was determined using TPR thermograms by Kissinger equation $(Eq. 2)$ [21]:

$$
\frac{-\ln\beta}{T_{\text{Max}}^2} = \frac{Ed}{RT_{\text{Max}}} - \frac{\ln AR}{Ed}
$$
 (2)

Where, β is the heating rate in °C/min, T_{Mav} is the highest thermal peak. Ed is the activation energy, R is the gas constant, and A is a pre-parameter value. A plot of 1000/ T_{Max} versus Ln β/T_{max}^2 is a straight line and E_d is obtained from the graph slope. In this method, the activation energy is calculated without the need for the identification of the mechanism and the degree of plicable to thermally stable materials such as pigments. chemical reaction. This method of measurement is apmetal oxides, nano-composites $[2.22.23]$. The TPR is a relatively new method for the characterization of solids and is one of the most widely used quantitative techniques for investigating the reduction behavior of catalysts [24]. In this study, the TPR was studied in the temperature range of 1 to 900 $^{\circ}$ C. Fig. 3 (a-d) shows the TPR analysis of $CuMn₂O₄$ nanostructures at three heating rates of 10, 15 and 20° C/min and Kissinger equation, respectively. In this experiment, a TPR/TPD device (SWORD Nano factory) and Kissinger equation were used to measure the activation energy. Thus, a mixture of H_2/Ar passes through the solid sample of CuMn₂O₄ nanostructures (0.30 g) and reacts with the structural oxygen of the sample during the heating operation. At the same time, the sample temperature rises at a linear rate. Due to the use of hydrogen in this ing the reduction rate of the test sample. In Fig. 3 (a), process, a signal is recorded from the device, indicattures. The highest thermal peak at 425° C indicates the ing rate of 10° C/min, indicating very pure nanostructhe sample shows no reduction below 425° C at a heathighest reaction rate at heating rate of 10° C/min.

Fig. 3. TPR analysis of CuMn₂O₄ nanostructures at (a) 10°C/min, (b) 15°C/min, (c) 20°C/min and (d) Kissinger Equation and activation energy of CuMn ${_{2}O_{_{4}}}$ nanostructures.

Table 1 summarizes the data of the highest thermal peaks indicating the highest reaction rate at all three heating rates. Also, T_{Max} 1000 versus Ln β/T_{Max}^2 is ated a straight line. The slope was calculated and the placed in the Kissinger equation (Eq. 3), which creactivation energy value was obtained 28.1 kJ/mol. Also, the activation energy value was 61.2 kJ/mol for Mn_2O_3 and 18 kJ/mol for CuO. A comparison of these tion energy is approximately the intermediate between three measured energy values showed that the activathe two other activation energy values $[25, 26]$.

$$
\ln\left(\frac{\beta}{T_{\text{Max}}^2}\right) = K + \frac{\frac{E_a}{R}}{T_{\text{Max}}}
$$
\n(3)

Aagnetic characterization of CuMn₂O₄ nanostruc-
tures

Studies on copper oxide (CuO) nanoneedle using hy-

Table 1. Summary of the highest thermal peaks of CuMn₂O₄ nanostructures according to Fig. 3 (a-c).

TPR Analysis	B (\degree C/min)	$\sim_{\text{Mas}}^{\text{S}}$
	15	430
	20	432

drothermal method have been shown ferromagnetic

e properties. The coercivity of CuO nanoneedles at 3 K

in Oe, was estimated to be 42 Ω [27, 28]. Fig. 4 shows

a VSM analysis of the hysteresis curve of CuMn₂O₄

na properties. The coercivity of CuO nanoneedles at 3 K in Oe, was estimated to be 42 Ω [27, 28]. Fig. 4 shows a VSM analysis of the hysteresis curve of $CuMn₂O₄$ sults show the moderate ferromagnetic properties of nanostructures after heat treatment at 600°C. The rethese nanostructures. According to the M-H (magnetic field - magnetic charge) diagram, the saturation magnetization (M_s) is equal to 0.7 emu/g, the magnetic field - magnetic charge) diagram, the saturation magremanence (M_r) is approximately 0.1 emu/g, and the coercivity (H_c) is approximately 330 Oe. In addition,

Fig. 4. Hysteresis curve of CuMn₂O₄ nanostructures cal-
cined at 600 °C for 4 h.

tence of a hysteresis loop or Mr, the results show the according to the data from this diagram and the exismoderate ferromagnetic properties of CuMn₂O₄ nano-
structures.

LCR test: Inductance, capacitance and resistance *analysis*

The LCR meter is used to measure the properties of pacitance, quality factor, electrical conductivity, and a dielectric material $[29]$. Fig. 5 $(a-d)$ shows the caelectrical resistivity of $CuMn₂O₄$ nanostructures, re spectively at different frequencies. Fig. $5(a)$ shows the electrical capacity diagram at a voltage of 10 mV and at 40 different frequencies $(1 \text{ kHz} -1000 \text{ kHz})$ using the LCR meter. The capacitance is calculated in accordance with the Eq. 4:

$$
C = \frac{K_{\varepsilon_0 A}}{d} \tag{4}
$$

Where, K is the dielectric constant and varies for each material. Approximately, $K = 1$ for air and ε_0 is the vacuum permeability that is 8.85×10^{-12} F/m. A is the capacitance area in m^2 , d is the distance between the two capacitors plates in m and C is the capacitance in farad $[30]$. Empirical data show that the sample capacity at a constant voltage of 10 mV decreases with increasing in frequency. The highest electrical capacity cal capacity decreases sharply in the frequency range in frequency of 1 kHz is equal to 3 . Also, the electricreases with a regular trend with increasing frequency of 26-100 kHz. The electrical capacity gradually dein the range of $100-1000$ kHz, and the highest capacity reduction is in the frequency of 1000 kHz.

ity at a constant voltage of 10 mV decreases with in

creasing in frequency. The highest electrical capacit

in frequency of 1 kHz is equal to 3. Also, the electrical

cal capacity decreases sharply in the frequency rang
 Fig. $5(b)$ displays the diagram of the quality factor at a constant voltage of 10 mV at 40 frequencies from rect relationship between the increasing in frequency 1-1000 kHz using a LCR meter. The results show a dicreases with increasing frequency. The sample quality and the quality factor as the sample quality factor infactor increases uniformly in the frequency range of $1-420$ kHz. At frequencies of $420-600$ kHz, a slight increase and decrease is observed. The highest quality factor increases is in the frequency range of 948-1000 kHz. The results indicating that the nanostructures can be used in the electronics industry because of their high-quality factor.

tivity at a constant voltage of 10 mV at 29 frequencies Fig. $5(c)$ shows the diagram of the electrical conducfrom 200-1000 MHz using the LCR meter. The results show that the conductivity decreases with increasing

Fig. 5. (a) Capacitance, (b) quality factor, (c) electrical conductivity and (d) electrical resistivity of CuMn₂O₄ nanostructures.

frequency. The highest conductivity decrease was in tivity increase was in the frequency of 308 MHz equal the frequency of 1000 MHz and the highest conducto $70.7 \mu S$. In the frequency range of $400-1000 \mu$ MHz, the conductivity decreases at a relatively constant rate, indicating the semiconducting property of these nano-
structures.

tance at a constant voltage of 10 mV at 20 frequencies Fig. $5(d)$ presents the diagram of the electrical resisfrom 1000-10000 Hz using the LCR meter. The results quency of 2421 Hz. Electrical resistance decreases show that the highest electrical resistance is in the frecrease in electrical resistance is in the frequency of regularly with increasing frequency. The highest de-8000 Hz equal to 67 Ω .

CONCLUSION

In this study, the CuMn₂O₄ nanostructures was prepared by stearic acid sol-gel synthesis method. In the XRD pattern analysis, the mean size of CuMn_2O_4 nanostruc tures was 28.1 kj/mol using TPR analysis, Kissinger tion. The activation energy value of these nanostructures was measured to be 23.03 nm by Scherrer equaequation and obtaining a linear slope of equation. The magnetic property of these nanostructures. In the LCR results of VSM analysis showed the moderate ferroanalysis, the highest electrical capacity was 3μ F in the frequency of 1 kHz and the highest quality factor was observed with increasing frequency in the range of 948-1000 kHz. The highest conductivity increase was in the frequency of 308 MHz equal to $70.7 \mu S$. In addition, the highest decrease in electrical resistance dicating the semiconducting property of these nanostructures. Therefore, $CuMn₂O₄$ nanostructures can be dicating the semiconducting property of these nanowas in the frequency of 8000 Hz equal to 67 Ω , inused in the electronics and fuel cell industries.

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